

Synthesis and Characterization of TiO₂ Nanorod Supported on Stainless Steel Mesh Prepared Via Sol-Gel Method

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ABSTRACT

Semiconductor metal oxide such as TiO₂ has received significant attention in the field of material science owing to its low cost, stability, environmental friendliness and excellent photocatalytic activity. However, fast electron-hole recombination rate, high band gap, post-separation and recovery of the used TiO₂ photocatalysts from wastewater after treatment limited its full scale industrial applications. In the light of this background, the present study investigated the synthesis of TiO₂ nanorods supported on stainless steel mesh using a sol-gel solution of Polyvinylpyrrolidone (PVP), ethanol and titanium tetrachloride (TiCl₄). The influence of pyrolysis temperature on the particles size, morphologies, microstructures and surface areas on the prepared supported catalyst were investigated. The as-synthesised nanorods were characterized by High Resolution Scanning Electron Microscope (HRSEM), High Resolution Transmission Electron Microscope (HRTEM), Energy Diffraction Spectroscopy (EDS), X-ray Diffraction (XRD), Brunauer-Emmett-Teller (BET) surface area, and X-ray photoelectron spectroscopy (XPS) analysis. The HRSEM and HRTEM analysis confirmed the formation of densely grown, well defined and uniformly aligned nanorod covered the whole surface of stainless steel mesh. The XRD analysis demonstrated the formation of pure tetragonal anatase phase of TiO₂ irrespective of pyrolysis temperature and no rutile or brookite phase were detected. The average crystalline size of TiO₂ nanorod decreased from 45 nm to 28 nm as the temperature increases from 300°C to 400 °C at constant holding time of 2 hrs. The specific surface area of the supported as-synthesised TiO₂ nanorods was in the order of prepared at 400 °C (81 m²/g) > 350 °C (77 m²/g) > 300 °C (62 m²/g). The XPS analysis revealed the Ti in the prepared TiO₂ is in +4 electronic state. This study has demonstrated that PVP/ethanol/TiCl₄ offers a facile, robust sol-gel related route for preparing supported TiO₂ nanorod.

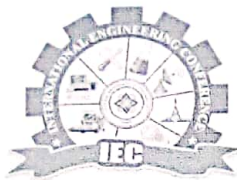
Keywords: Sol-gel method, TiO₂ nanorod, synthesis, stainless steel mesh

1 INTRODUCTION

In the last couple of years, semiconductor metal oxides such as ZnO, TiO₂, WO₃, Fe₂O₃, ZnS, CdS, GaP, ZrTiO₄, and SiO₂ have been utilized for different purposes in the field of the photocatalytic technology. Among the aforementioned heterogeneous metal oxides, TiO₂ photocatalysts has received considerable attention in the field of material science due to its low cost, photochemical stability, environmental friendliness and exceptional photocatalytic behaviour (Liu et al., 2016; Shinde et al., 2015; Zhao et al., 2017). Titanium (IV) oxide can be found in the natural form of titanic anhydride or titanic acid anhydride and exists in three major phases namely rutile, anatase and brookite (Álvarez et al., 2010; Gharakhlou and Sarvi, 2017). And among the phases, anatase remains the most preferred owing to its photocatalytic and photochemical stability. Titanium precursor such as TiCl₄ or titanium alkoxides are mostly used in the preparation of anatase TiO₂.

However, high electron-hole recombination rate from valence band to conduction band contributed to the low photosensitivity and photocatalytic efficiency of TiO₂

(Bestetti et al., 2010; Gaya and Abdullah, 2008; Tijani et al., 2017). Also, high capital and running cost associated with the post-separation and recovery of the used TiO₂ photocatalysts from wastewater after treatment limits its full scale industrial applications (Hintsho et al., 2014; Jiang et al., 2017). In view of the highlighted shortcomings, different approaches have been adopted to overcome the challenges. One of such strategies is the doping or immobilisation of TiO₂ nanoparticle with metals and non-metals impurities, semiconductor coupling or sometimes fluorination process (Tijani et al., 2017). Doping creates a defect level within the band gap causing reduction in the electron excitation of TiO₂ and possibly extends the absorption threshold to the visible region and enhanced the photocatalytic activity (Kuriechen and Murugesan, 2013; Lee et al., 2015; Nunes et al., 2017). Supported or unsupported Nano TiO₂ materials in the form of powder, crystals, rods, or thin-films have been prepared via different methods namely; electrochemical, magnetic sputtering, photo deposition, sulfate process, chloride process, solvo-thermal, electron beam evaporation, precipitation, hydrothermal, chemical vapour deposition, thermal evaporation, wet impregnation dip coating, and sol-gel process (Carlucci et al., 2014; Chong



et al., 2010; Jiang et al., 2017). Among the aforementioned techniques, sol-gel technique appeared most suitable due to ease of manipulation of synthesis conditions resulting to the formation of nanoparticles with good homogeneity and precise phase structure (Monreal et al., 2009; Nainani et al., 2012; Wahyuningsih, et al., 2017).

Furthermore, researchers have developed TiO₂ nanomaterials on different fixed supports such as zeolite, clay, activated carbon, glass fiber, quartz, stainless steel mesh amongst others in order to solve the post-filtration of TiO₂ particles after treatment (Omri et al., 2015). Till date, efforts are still ongoing towards finding a suitable and stable substrate to support TiO₂-based photocatalyst and only very few reports exist on the synthesis of supported TiO₂ nanorods (Zhang et al., 2017). For instance, Lia et al., (2009) prepared crystalline TiO₂ nanorod array on the pre-treated glass substrate via hydrothermal route and found that the well-aligned single crystal of TiO₂ grown on pre-treated glass substrate exist as a rutile polymorph with an average diameter and length of approximately 21 and 400 nm, respectively. Mahendiran et al., (2014) prepared unsupported TiO₂ nanorods via a novel direct sol filling and heating sol-gel template method involving titanium (IV) isopropoxide as a precursor. The authors found that the as-prepared TiO₂ nanorod exhibited pure anatase polymorph with an average size of 10 nm and 99% of Ti existed in +4 oxidation state. Shinde et al., (2015) reported the hydrothermal synthesis of TiO₂ nanorods on Transparent Conducting Oxide (TCO) and found that the deposited TiO₂ nanorods were purely rutile phase with tetragonal crystal structure.

In this present work, stainless steel mesh supported TiO₂ nanorods were synthesized using sol-gel solution comprises Polyvinylpyrrolidone (PVP), ethanol and titanium tetrachloride (TiCl₄). The influence of pyrolysis temperature on the size and shape of the nanorods were investigated. The prepared supported TiO₂ nanorods were characterised for their morphologies, microstructures, elemental composition, mineralogical phases, surface areas, and surface chemical states using High Resolution Scanning Electron Microscope (HRSEM), High Resolution Transmission Electron Microscope (HRTEM), Energy dispersive X-ray spectroscopy (EDS), X-ray diffraction (XRD), Brunauer–Emmett–Teller (BET) measurements, and X-ray photoelectron spectroscopy (XPS).

2 METHODOLOGY

2.1. MATERIALS

Titanium (IV) chloride with (0.09 M in 20% HCl) and Polyvinylpyrrolidone (500 g) powder were obtained from Sigma Aldrich. While ethanol was supplied by B & M Scientific. The chemicals were of analytical grade and used without any further purification.

2.2 SYNTHESIS OF TiO₂ NANOROD SUPPORTED ON STAINLESS STEEL MESH

Twenty (20) gram Polyvinylpyrrolidone (PVP) was added to 40 mL ethanol in a beaker and stirred on a hot plate until the PVP completely dissolved and a yellowish transparent gel was obtained. 10 mL titanium (IV) chloride solution was slowly added to 20 mL solution containing PVP and ethanol. The mixture was sealed in a bottle and then stirred on a magnetic stirrer at room temperature (25 °C) for 3 hrs. At the end of 3 hrs stirring, a slightly transparent PVP/TiCl₄ viscous solution was obtained. This step was conducted in a ventilated fume hood. During the addition of 10 mL of 0.09 M TiCl₄, the transparent solution rapidly became a brownish yellow in colour. The addition of 6 mL of 0.09 M TiCl₄ was accompanied by the release of white fumes of hydrochloric acid, and the stirring was continued for 15 min until the white fumes disappeared. The formed viscous solution was continually stirred until the white fumes disappeared.

Before manual coating of the stainless steel mesh with sol-gel, a 4 × 4 cm stainless steel mesh was rinsed with acetone, ethanol and water, and then dried in an oven at 90 °C for 5 min in order to remove organic and inorganic contaminants. The cleaned stainless steel mesh was weighed and coated manually with the prepared sol-gel formulation of PVP/TiCl₄. The sol-gel solution was coated on 4 × 4 cm stainless steel mesh using the dip-coating method with a drawing speed of 5.0 cm/min. This procedure was repeated three times and thereafter the coated mesh was air dried and later pyrolysed in the furnace at different temperature 300 °C, 350 °C and 400 °C. A heating rate of 50 °C/min and holding time of 2 hrs was adopted. The procedure also included a flow of nitrogen gas through the inner tube at 20 mL/min in order to create an inert environment and to prevent side reactions such as hydrolysis. The synthesized supported TiO₂ nanorods prepared at different temperature 300 °C, 350 °C and 400 °C for 2 hrs holding time was coded JT1, JT2 and JT3.

2.3. CHARACTERIZATION OF SUPPORTED TiO₂ NANOROD

Philips X-pert pro MPD X-ray diffractometer (XRD) Cu-K radiation at 40 kV and 40 mA was used to determine the phase composition and estimation of the crystal size of the synthesised catalysts. The X-ray patterns were scanned in the 2θ range from 20-80 with a step 0.050, counting time 1s/step. The crystalline size was calculated from the peak full-width at half maximum with corrections for instrument line broadening by using the Scherer equation, $D = k \lambda / (\beta \cos \theta)$, $K = 0.89$, $\lambda = 0.1541$ nm, θ is the half-diffraction angle, β the half-peak width, D the diameter of crystalline particle. HRSEM equipped with EDS was used to determine the particle morphology and elemental composition of the synthesised catalysts. Microstructures and particle distribution patterns were

analyzed by a Zeiss Auriga High resolution transmission electron microscope (HRTEM). Approximately 0.02 g of the synthesised products was suspended in 10 mL methanol and thereafter subjected to ultra-sonication until complete dispersion was achieved. One or two drops of the slurry was dropped onto a holey carbon grid with the aid of a micropipette and subsequently dried via exposure to photo light. For BET N₂ adsorption, about 100 mg of the dry powder sample in a sample tube, was first degassed at 90°C for 4 hours to remove residual water and other volatile components that were likely to block the pores. The BET surface area and average pore volume distributions were obtained from the plot of volume adsorbed (cm³/g STP) versus relative pressure. The N₂ adsorption-desorption isotherms were collected at -196° C using Micromeritics ASAP 2020 Accelerated Surface Area and Porosimetry analyser. A XPS PHI 5400 equipped with hemispherical sector analyser operated using non-monochromated Al K α x-rays with energy 1486.6 eV, at 300 W and 15 kV was used to examine the surface composition of the material. The energy scale was calibrated using the Au 4f_{7/2} at 83.95 eV and the linearity of the scan was adjusted to measure the Cu 2p_{3/2} at a position of 932.63 eV. Surveys were scanned (2.5 eV/s) with pass energy of 178 eV and detailed spectra were collected with pass energy of 44 eV and scan rate of 0.625 eV/s. The photoelectron take-off angle for all measurements was 45°. All spectra obtained were energy corrected using the aliphatic adventitious hydrocarbon C (1s) peak at 284.8 eV. The XPS Peak 4.1 software was used for data analysis and fits.

3 RESULTS AND DISCUSSION

High resolution scanning electron microscopy (HRSEM) was used to examine the morphology of the prepared supported TiO₂ nanorods. The HRSEM micrographs of the titania nanorods at different temperatures (300 – 400 °C) for 2 hrs holding time are depicted in Fig. 1. As shown in Fig. 1(a), there was no formation of nanorods on the stainless steel mesh, when the pyrolysed temperature was 300 °C (Fig 1 (b)), instead agglomerated irregular nanocrystals and nanorods that stick to the stainless steel mesh were observed. This implies that the polymer precursor (PVP) is not stable at temperature beyond 200 °C, thus initiation of crystallization of the TiCl₄ precursor into TiO₂ nanorods occurred. The length of the nanocrystals and nanorods according to imageJ software were found to be 10 nm and 16.8 nm respectively. As the pyrolysed temperature increases to 350 °C (Fig. 1 (c)) uniformly distributed, well aligned, densely agglomerated tetragonal bunches-like morphology were observed. The length of the nanorods increased to 35 nm at 350 °C for 2 hrs holding time and a complete disappearance of the nanocrystals was observed. This suggests thermal conversion of the PAN into carbonaceous species during the heating period. With further increase in temperature to 400 °C, short, homogeneous, completely grown, regularly

aligned and less aggregated nanorods resembling rice grain were observed. With increasing the pyrolysis temperature, the length of the nanorods slightly reduced to 26 nm. This was an evidence of a fast nucleation process responsible for the formation of densely grown, well defined and uniformly aligned nanorod. The reduction in the size of the nanorods is an indication that the as-synthesised supported TiO₂ nanorod flowers may have a large surface area and exhibit high photocatalytic activity under the incident photons (Omri et al., 2015). It is worthwhile to note, that PAN and its decomposition products acts as an adhesive that stick the TiO₂ nanorods together onto the stainless steel mesh. It is evident that the morphological evolution that is the shape and size of the nanorod is strongly a temperature dependent phenomenon.

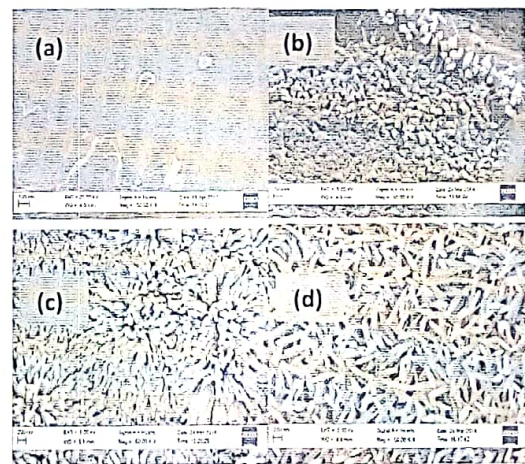


Fig. 1: HRSEM micrographs of (a) stainless steel mesh alone (b) titania nanorods prepared at 300 °C (c) titania nanorods supported prepared at 350 °C (d) titania nanorods supported on stainless steel mesh deposited at 400 °C for 2 hrs

HRTEM ANALYSIS

HRTEM was utilized to further examine the detailed microstructures of the supported TiO₂ nanorods prepared by pyrolysis at 300°, 350°, or 400° C for 2 hrs. The HRTEM micrographs of the supported TiO₂ nanorods pyrolyzed at 300 °C, 350 °C or 400 °C in N₂ atmosphere at 20 mL/min for 2 hrs is represented in Fig. 2. Fig. 2 shows the HRTEM images of supported TiO₂ nanorods prepared by pyrolyzed in N₂ atmosphere at 300 °C for 2 hrs. In Fig. 2 (a), a mixture of nanocrystals and nanorods were formed, which means that some of the TiCl₄ precursor was still embedded in the gradually decomposing PVP polymer. With increase in temperature to 350 °C for 2 hr holding time, the initially observed mixed nanocrystals and nanorods completely transformed to a full, thick,

smooth and orderly arranged nanorods (Liu et al., 2016). This clearly shows complete decomposition of PVP and non-entrapment of $TiCl_4$ precursor. Furthermore, the length of the nanorod was determined using imageJ and found to be in the range of 35 to 40 nm. When the temperature was increased to 400 °C hr (Fig.2 c), a well resolved, thin and highly crystalline TiO_2 nanorods was homogeneously deposited on the stainless steel mesh.

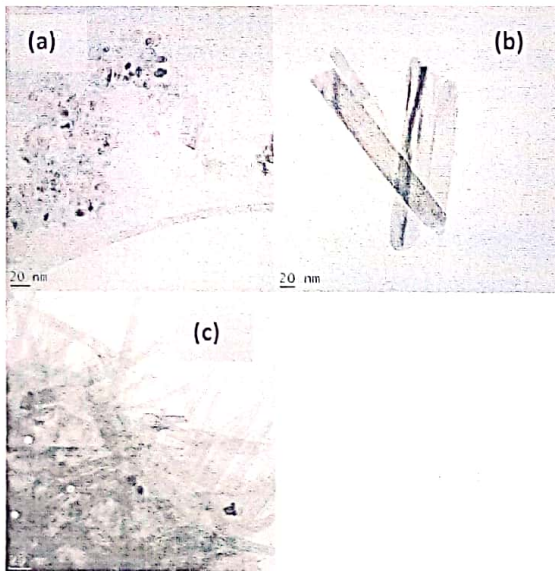


Fig. 2: HRTEM micrographs of (a) titania nanorods supported on stainless steel mesh deposited at 300 °C (b) titania nanorods supported on stainless steel mesh deposited at 350 °C (c) titania nanorods supported on stainless steel mesh deposited 400 °C for 2 hrs

EDS ANALYSIS

An elemental composition analysis of the stainless steel mesh and that of the as-synthesised supported TiO_2 nanorods were determined using EDS. Fig. 3 (a) and (b) show the EDS spectra of stainless steel mesh alone and supported TiO_2 nanorods. The EDS spectrum of stainless steel mesh recorded in the binding energy region of 0–20 keV and the following individual elements C, Cr, Fe, Ni, Si, Mo were detected in different proportion.

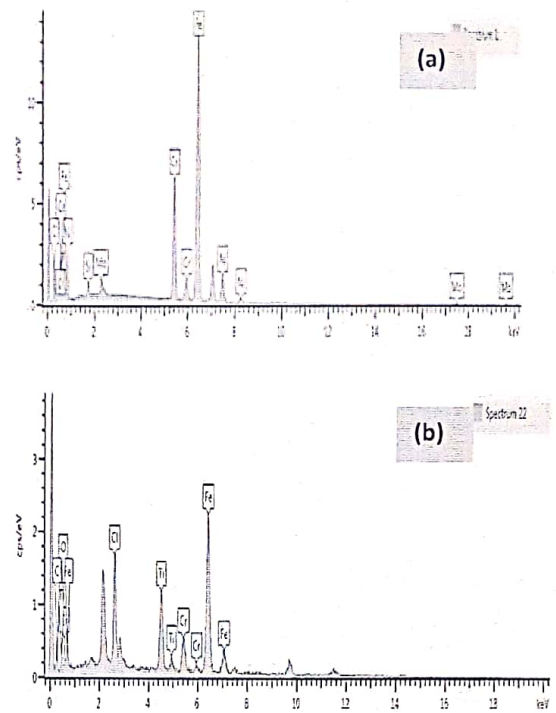


Fig. 3: EDS spectrum of (a) ordinary stainless steel mesh (b) TiO_2 nanorods supported on stainless steel mesh

However, in Fig. 3 (b), two prominent signals different from Fig 3 (a) were noticed in the binding energy region of 4.508 and 0.525 keV respectively, which corresponds to Ti and O were identified. This means that the as-synthesised catalyst consist Ti and O with a near atomic ratio of 1:2 which is closely agrees with the stoichiometry of TiO_2 . The atomic percentages of Ti and O in the prepared samples are approximately 36.21% and 22.80% respectively. The elements such as Fe and Cr originated from the stainless steel mesh while the residual chlorine is from the titanium precursor ($TiCl_4$).

XRD ANALYSIS

XRD analysis was done to investigate crystallite size and crystal structure of the TiO_2 nanorods. Fig. 4 reveals the influence of pyrolysis temperature on the crystalline nature of the as-synthesised materials. XRD pattern in Fig. 4 (a) exhibited weak diffraction peaks at the 2 theta value of 58.2, 59.2 and 66.1, which correspond to Fe, Cr and Si respectively. Besides these elements, it can be seen from the XRD patterns especially Fig. 4 (b-d) the presence of the following strong broad diffraction peaks at 25.55°, 38.3°, 47.5°, 53.60°, 63.2°, 69.7°, and 75.2° irrespective of the pyrolysed temperature. These diffraction peaks indicate the presence of following crystal planes (101), (004), (200), (211), (204), (220), and (215).

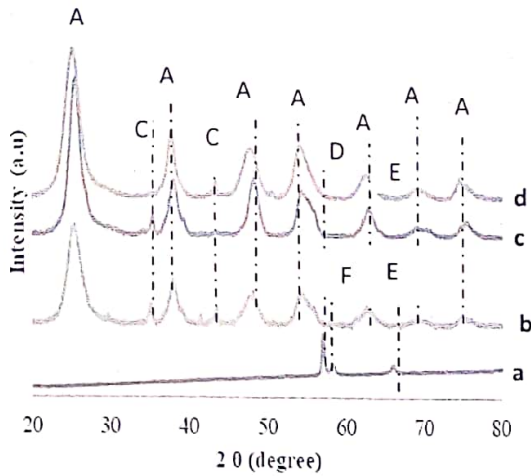


Fig. 4: XRD pattern of the catalysts: (a) stainless steel mesh alone (b) (300 °C for 2 hrs) (c) (350 °C for 2 hrs) (d) (400 °C for 2 hrs) at constant heating rate of 50 °C/min A=anatase, C=carbon, D=Fe, F=Cr E=Si

All diffraction peaks closely agree with the standard (JCPDS: No 21-1272). The strong peak clearly seen from Fig. 4 (b-d), confirmed the formation of highly crystalline TiO₂ products. For each temperature, the XRD patterns showed the formation of pure the tetragonal crystal anatase polymorph of TiO₂ nanorods and no peaks correspond to rutile or brookite phases at each temperature were noticed. Other diffraction peaks observed in Fig 4 (b-c) at 2 theta value of 35.4° (002), 41.2° (111) and 45.4° (100) are characteristics peaks of a typical carbon species derived from PVP. Carrenõ et al., (2008) shown that during the pyrolysis reaction, polymer precursor often decomposed to residual carbon and possible replacement of oxygen in TiO₂ by carbonaceous materials from PVP precursor and Ti-O-C bond possibly formed (see Fig.8). It was observed that as the pyrolysis temperature increases the carbon peak disappeared, an indication of partial or complete decomposition of PVP and no further encapsulation or entrapment of the nanorods within the polymer precursor chain. This further suggests that as the temperature increases, there was a simultaneous pyrolysis of PVP followed by oxidation of TiCl₄ precursor into TiO₂ nanorods. The average size of the nanorods were determined using using the Debye-Scherrer equation shown in equation 1, where the 2 theta value of the intense anatase peak (101) in the synthesized TiO₂ nanorods was selected.

$$d = \frac{k\lambda}{\beta \cos\theta}$$

.....(1)
where d in equation 1 represents the crystallite size in nanometer, K = 0.94, λ is the wavelength and equal to 0.1541 nm, θ is the half-diffraction angle, β is the full width at half-maximum in radian of 2 θ value (25.55°). According to equation 1, the average crystalline size of TiO₂ nanorod decreased from 45 nm to 28 nm as the temperature increases from 300°C to 400 °C at constant holding time of 2 hrs. The slight broad nature of the diffraction peaks at 2 theta value of 25.3° may be responsible small size of the nanorods.

BET ANALYSIS

The N₂ adsorption-desorption isotherms of the TiO₂ nanorods supported on stainless steel mesh pyrolysed in the furnace at 300 °C, 350 °C and 400 °C for 2 hrs are shown in Figure 5.

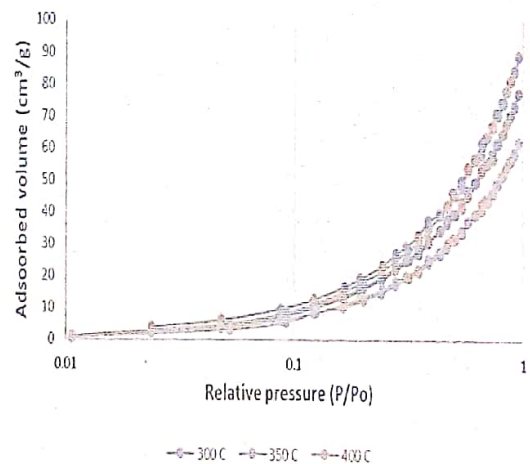


Fig.5: BET surface area of the prepared supported TiO₂ nanorods at different temperature

The specific surface areas and pore volumes determined using Micrometrics ASAP 2020 surface area and Porosity Analyzer are shown in Table 1. According to Table 1, the BET specific surface areas of the as-prepared supported TiO₂ nanorods at 300 °C, 350 °C and 400 °C pyrolysis temperature and constant holding time of 2 hrs are 62, 77 and 81 m²/g. It is evident from Table 1 that, as the temperature increases the BET surface area also increases, which is attributable to the thermal decomposition of PVP into carbonaceous species and incorporation of such carbon onto the lattice crystals of the supported titania rods.

TABLE 1: SPECIFIC SURFACE AREA AND PORE VOLUME OF AS-SYNTHESISED SUPPORTED TiO₂ NANORODS COMPARED TO THE COMMERCIAL TiO₂ POWDER (DEGUSSA P25)

Samples	BET surface area (m ² /g)	Pore volume (cm ³ /g)
P25 (commercial TiO ₂)	56	0.64
300°C for 2 hr	62	1.12
350°C for 2 hr	77	1.21
400°C for 2 hr	81	1.35

It was also noticed that the surface area of the as-synthesised TiO₂ nanorods is greater than that of commercial TiO₂ nanoparticles. The possible reason for the increased specific surface area might be attributed to the pyrolysis temperature and the immobilisation effect of carbonaceous species from PVP, which perhaps lead to substitution of the oxygen in TiO₂ with carbon. According to the IUPAC classification, the as-synthesised supported TiO₂ nanorod exhibited type IV-isotherm and H₂ hysteresis respectively irrespective of the pyrolysis temperature. This is an indication that the synthesized samples are mesoporous in nature as evident in the sloping adsorption branch of loop and small pore volume. In addition, the presence of a relatively sharp steep desorption branch observed at high relative pressure (P/P₀) range also suggest the formation of mesoporous material with particles size in the range of 2 -50 nm.

XPS ANALYSIS

The XPS analysis was done to identify the possible chemical state or the surface valence states of elements present in the supported TiO₂ nanorods. Fig. 6 represents the High resolution XPS survey of the supported TiO₂ nanorods prepared at 400 °C for 2 hrs.

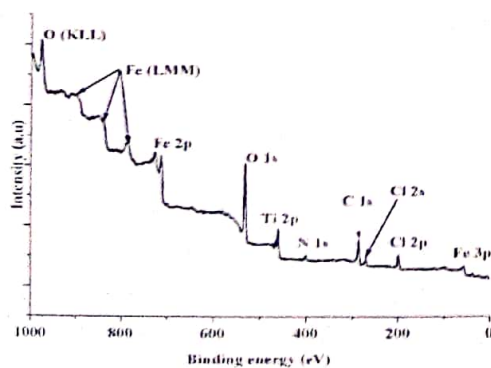


Fig. 6: General XPS survey of supported TiO₂ nanorods pyrolysed at 400 °C for 2 hrs

The general survey of the supported TiO₂ nanorods pyrolysed at 400 °C for 2 hrs revealed the presence of the following elements: Fe, O, Ti, N, C, and Cl. The Fe originated from the stainless steel mesh while Cl and N are from titanium and polymer precursor respectively. The binding energy regions of 459.12 and 530.6 eV correspond to Ti 2p and O 1s respectively. Furthermore, the XPS survey shown in Fig. 8 depict the Ti 2p and possessed two peaks in the binding energy region of 459.2 and 465.7 eV.

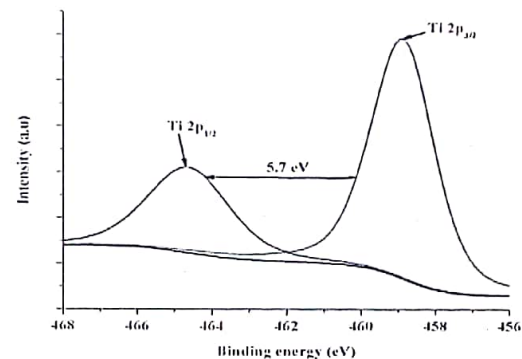


Fig. 7: XPS survey of Ti 2p envelop in TiO₂ nanorods

Also in Fig. 7, it can be noticed that there is an existence of one broad, sharp and narrow peaks at the binding region of 459.3 and 465 eV. This implies that the Ti in TiO₂ nanorods exist in +4 oxidation state. The sharp peak in the Ti 2p orbital connote Ti 2p_{3/2} while the broad peak represents Ti 2p_{1/2}. The presence of these two orbitals mean that Ti occurred in one chemical state. Moreso, the differences between the binding energies values of Ti 2p_{3/2} and Ti 2p_{1/2} was 5.7 eV, which further buttress the fact that Ti existed in +4 chemical state in the as-synthesised TiO₂ nanorods. The binding energies for Ti and O obtained in this study agree with the literature value reported for TiO₂ nanorods (Lia et al., 2009; Mahendiran et al., 2014; Shinde et al., 2015)

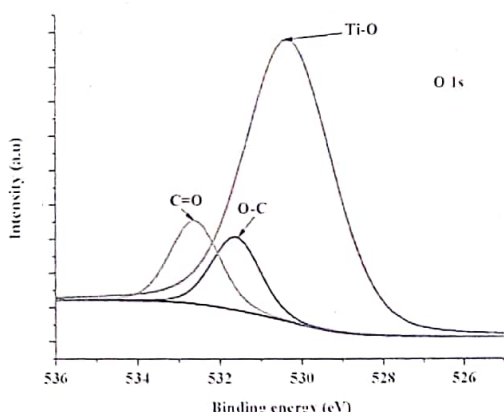


Fig. 8: The XPS spectrum of O 1s enveloped in TiO₂ nanorod synthesised at 400 °C for 2 hrs

Fig. 8 shows the XPS scan for O 1s, three prominent peaks in the binding energy region of 530.6, 531.7 and 532.9 eV were found and correspond to Ti-O, O-C and C=O bond respectively. Out of these three peaks, the most dominant can be found in the binding energy region of 530.6 eV. The presence of this peak further support the formation of Ti-O bond of TiO₂, an indication of titanium atom bonded to oxygen. The XPS spectra further complements the EDS result shown in Fig. 3.

4. CONCLUSION

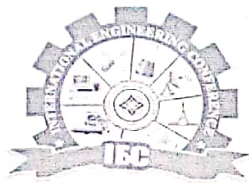
In this study, TiO₂ nanorods supported on stainless steel mesh was prepared via sol-gel methods followed by pyrolysis at different temperatures. Based on this investigation, the following conclusions were drawn. The HRSEM and HRTEM micrographs confirmed the formation of TiO₂ nanorods of different length at each synthesised temperature. The pyrolysis temperature influenced the phase, morphologies and the surface areas of the TiO₂ nanorods. The carbonaceous species obtained via the decomposition reaction of PVP in nitrogen atmosphere acts as dopant and adhesive respectively. The supported TiO₂ nanorods prepared by pyrolysis at 300 °C, 350 °C and 400 °C for 2 hrs was actually supported carbon doped TiO₂ nanocomposites. The average crystalline size of TiO₂ nanorod decreased from 45 nm to 28 nm as the temperature increases from 300°C to 400 °C at constant holding time of 2 hrs. Among all, the product obtained at 400 °C (2 hr) was found highly crystalline with highest porosity.

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