

**Non-Metallic Material Science** https://ojs.bilpublishing.com/index.php/nmms

ARTICLE

# High Energy X-Ray Dosimetry Using (ZnO)<sub>0.2</sub>(TeO<sub>2</sub>)<sub>0.8</sub> Thin Filmbased Real-time X-Ray Sensor

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# ABSTRACT

This study reports the dosimetric response of a  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor irradiated with high-energy X-ray radiation at various doses. The spray pyrolysis method was used for the film deposition on soda-lime glass substrate using zinc acetate dehydrate and tellurium dioxide powder as the starting precursors. The structural and morphological properties of the film were determined. The I-V characteristics measurements were performed during irradiation with a 6 MV X-ray beam from a Linac. The results revealed that the XRD pattern of the AS-deposited thin film is non-crystalline (amorphous) in nature. The FESEM image shows the non-uniform shape of nanoparticles agglomerated separately, and the EDX spectrum shows the presence of Te, Zn, and O in the film. The I-V characteristics measurements indicate that the current density increases linearly with X-ray doses (0-250 cGy) for all applied voltages (1-6 V). The sensitivity of the thin film sensor has been found to be in the range of 0.37-0.94 mA/cm<sup>2</sup>/Gy. The current-voltage measurement test for fading normalised in percentage to day 0 was found in the order of day 0 > day 15 > day 30 > day 1 > day 2. These results are expected to be beneficial for fabricating cheap and practical X-ray sensors.

Keywords: Thin film; X- ray radiation; I-V characteristics; Dosimetry

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#### ARTICLE INFO

Received: 27 December 2022 | Revised : 30 December 2022 | Accepted: 2 February 2023 | Published Online: 16 February 2023 DOI: https://doi.org/10.30564/nmms.v5i1.5369

CITATION

Idris, M.M., Olarinoye, I.O., Kolo, M.T., et al., 2023. High Energy X-Ray Dosimetry Using  $(ZnO)_{0.2}(TeO_2)_{0.8}$  Thin Film-based Real-time X-Ray Sensor. Non-Metallic Material Science. 5(1): 4-13. DOI: https://doi.org/10.30564/nmms.v5i1.5369

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# **1. Introduction**

The continuous expansion of ionising radiation applications in many areas has necessitated the use of dosimeters with special physical features, such as small size <sup>[1]</sup>. The quest to design miniaturised dosimeters with small active volumes has consequently led to increasing research into materials in thin film form for their dosimetric properties <sup>[2-4]</sup>. Numerous techniques of deposition, both physical and chemical, have been adopted for transparent conducting oxide thin film preparation. The desired properties of the prepared film are predetermined by the deposition technique used. These can greatly affect the functionality of the film device <sup>[5]</sup>.

The widespread usage of transparent conducting oxide (TCO) thin films in optoelectronic devices such as touch screens, liquid crystal displays, solar cells, and light-emitting diodes in recent years has drawn substantial scientific interest <sup>[6,7]</sup>. Owing to its strong electrical conductivity and excellent transparency to visible light, indium tin oxide (ITO) is the most widely used TCO <sup>[8]</sup>. The high price, limited supply, and toxicity of indium, the main component of ITO, have raised significant concerns about finding viable substitutes <sup>[9]</sup>. In addition, ITO loses some of its electrical and optical qualities when exposed to a hydrogen plasma environment <sup>[10]</sup>.

Due to its excellent optical and electrical characteristics as well as the low cost, non-toxicity, and abundance of Zn, ZnO is a particularly alluring material in this respect and a great substitute material for ITO <sup>[11]</sup>. At ambient temperature, ZnO is an n-type semiconductor material with a relatively broad band gap energy of 3.37 eV and a significant exciton-binding energy (60 meV) <sup>[12]</sup>. High thermal stability, strong stability in hydrogen plasma, and high electrochemical stability are some of its additional advantageous characteristics when compared to ITO <sup>[11,13]</sup>.

Tellurium oxide is a p-type semiconductor material with a band gap of about 2.88 eV and interesting non-linear optical properties <sup>[14]</sup>. It is characterised by a high dielectric constant in both crystal and film making it a potential candidate for future use in ultrahigh integrated electronic devices <sup>[15]</sup>. TeO<sub>2</sub> based materials have attracted considerable research interest in recent years due to their high refractive index, good non-linear optical properties and electrical semi-conductivity appealing for various applications <sup>[16]</sup>.

Transparent conducting oxide has been reported to be deposited on substrates using a variety of deposition techniques, including pulse laser deposition (PLD) <sup>[17]</sup>, chemical vapour deposition (CVD) <sup>[18]</sup>, radio frequency (RF) sputtering <sup>[19]</sup>, and the sol-gel method <sup>[20]</sup>. RF sputtering method and convoluted deposition techniques, like MBE, showed good thin film defect density, crystal structure consistency, and high deposition speeds with little defect concentration <sup>[19]</sup>. The deposition technique, however, costs more than conventional deposition techniques while producing high deposition rates. The spray pyrolysis method can produce high-quality thin films on substrates at moderate temperatures with a low cost of operation <sup>[21-31]</sup>.

With regard to the lattice characteristics and thermal mismatching between the substrate and the film owing to the development of stress in the deposited films, the choice of substrate material is crucial for the formation of ZnO doped TeO<sub>2</sub> thin films <sup>[24]</sup>. Aside from that, the type of substrate can also affect how ZnO film nucleates and grows. Soda-lime glass<sup>[25]</sup>, sapphire (Al<sub>2</sub>O<sub>3</sub>)  $^{[26]}$ , Si (1 0 0)  $^{[9]}$ , and GaAs  $^{[27]}$  are examples of substrates often employed for the deposition of a most metal oxide including ZnO and TeO<sub>2</sub> thin films. Among the substrate materials, soda-lime glass substrate has been proven suitable [29-36]. In this study, we prepared a  $(ZnO)_{0,2}(TeO_2)_{0,8}$  thin film sensor by spray pyrolysis with the aim of testing the dosimetric response during irradiation with high-energy x-rays.

Thin film sensor undergoes structural changes when exposed to ionising radiation such as optical, structural and electrical properties <sup>[29]</sup>. Ionizing radiation can cause changes in the properties of the thin film which depend on the radiation dose, parameters associated with the films, and the radiation type <sup>[7,30-32]</sup>. Previous studies have investigated materials in thin film form as radiation sensors for conceivable dosimeter designs <sup>[29-37]</sup>. For gamma sens-

ing and dosimetry application, In<sub>2</sub>O<sub>3</sub> thin film was recently deposited using the spray pyrolysis method and optimised <sup>[33]</sup>. It was discovered that the deposition conditions affected the film's gamma-ray sensitivity. The investigation came to the conclusion that the thin films are suitable for use as photon dosimeters. However, the toxic nature of indium oxide [34] could prevent the widespread manufacture of dosimeters based on the substance. Another study <sup>[35]</sup> discovered that polymer-encased CS<sub>4</sub>PbI<sub>6</sub> thin films were extremely sensitive to X-rays, reaching 256.20 cu cm<sup>-2</sup> for 30 keV X-rays at a bias voltage of 10 V. The materials were determined to be reliable, strong, and reusable for radiation measurement. The effects of gamma photon irradiation on the crystallinity, microstructure, optical transmission, and photoluminescence characteristics of PbIn thin films deposited by the spin coating process were also reported by Aldawood et al.<sup>[35]</sup>. The examined film's characteristics were discovered to change with photon exposure and may be calibrated for radiation monitoring and detection. Other thin film materials, including ZTO [37],  $ZrO_2$   $^{[36]},$   $MoO_3$   $^{[38]},$  ZnO  $^{[39]},$  ITO  $^{[40]},$   $TeO_2$   $^{[41]},$  and others, have shown properties that make them highly sensitive to radiation and, as a result, appropriate for the fabrication of real-time dosimeters.

When subjected to photons, ZnO and TeO<sub>2</sub> have each individually demonstrated changes in their optical, structural, and electrical characteristics <sup>[16,39,41]</sup>. These findings show that both materials are highly photon sensitive due to their electrical and optical characteristics. However, it could be possible to maximise radiation sensitivity by combining both materials in thin film form. As a result, for the first time, this study will look at how the electrical characteristics of  $(ZnO)_{0.2}(TeO_2)_{0.8}$  composite thin films are altered for X-ray photon detection and dosimetry. The fabrication of thin-film-based miniaturized dosimeters that is extremely sensitive, stable, repeatable, reusable, and capable of real-time readout in radiation-related applications such as medicine was the motivating factor behind this study. This study aims to investigate and evaluate the effects of various X-ray doses on the current-voltage (I-V) properties of the  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor, as well as the sensitivity and lowest detectable dosage at various bias voltages.

# 2. Experimental procedure

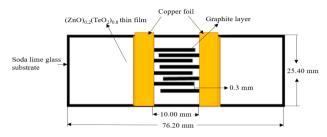
#### 2.1 Chemical synthesis

The zinc acetate dehydrate was slowly added (drop-wise) into a beaker placed on an analytical balance until 0.863 g was measured, then 2.0 mL of acetyl acetone and 58.0 mL of methanol were added, respectively. The mixture was stirred vigorously using a magnetic stirrer for about 15 minutes at room temperature until the solute completely dissolved. Also, 0.638 g of tellurium dioxide solute with a molecular weight of 159.6 g/mol was measured and then dissolved in 40 mL of HCl (43 mol% concentration) and 20 mL of methanol, respectively. The solution was vigorously mixed while being heated at 60 °C for about 15 minutes using a magnetic stirrer hotplate, until the solute dissolved completely. The addition of methanol to the tellurium dioxide solution was to prevent precipitation.

#### 2.2 Thin film deposition

Spray pyrolysis was used to create the (ZnO)<sub>0.2</sub>(TeO<sub>2</sub>)<sub>0.8</sub> thin film sensors. Compressed air was employed as a carrier gas to spray the atomized precursor solution of ZnO-doped TeO<sub>2</sub> onto a heated soda-lime glass substrate using a desktop-style automated ultrasonic spray pyrolysis coating apparatus (U-spray USP 1500). 10 mL of the precursor solution was placed in the dispensing tank of the depositor, and the substrate temperature was maintained at 300 °C throughout the deposition procedure. The flow rates of the solution, air, and the distance from the nozzle to the substrate are 0.15 mL $\cdot$ min<sup>-1</sup>, 0.2 kg $\cdot$ cm<sup>-2</sup>, and 3.0 cm, respectively. An innate solution tube and a glass tube, through which carrier gases travel, make up the spray nozzle. The solution is automatically sucked in and then sprayed when pressure is applied to the carrier gas, which creates a vacuum at the nozzle's tip. A thin film of thickness 375 nm was created

using 1.2 mL of the synthesised  $(ZnO)_{0.2}$   $(TeO_2)_{0.8}$ solution that had been atomized. On the thin film that had been created, a thick film of interdigitated graphite electrodes with an equal inter-electrode spacing of 0.3 mm was printed, together with two copper foils positioned at the margins of the graphite electrodes 10 mm apart from one another. **Figure 1** depicts the schematics diagram of the  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor that has been constructed.



**Figure 1.** Schematic diagram of the prepared  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor.

#### 2.3 Characterisation

The film crystallinity and crystalline phase of  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film was analyzed by X-ray diffraction measurement which was carried out at room temperature by using PHILIPS (PW 3040/60 MPD X'PERT HIGH PRO PANALYTICAL) diffractometer system (scan step of 0.05° (20), counting time of 10.16 s per data point) operated at 40 kV and 30 mA. It is equipped with a Cu tube for generating Cu-K $\alpha$  radiation (k = 1.5406 Å); as an incident beam in the 2-theta mode over the range of 10°-80°. The morphological analysis was performed by FEI (Nova Nanosem 230) field emission scanning electron microscope (FESEM). Energy-dispersive spectrum (EDX) analysis of the film was performed during FESEM measurements.

# 2.4 Irradiation and I-V characteristic determination

The  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor was exposed to various doses (dose rate) of 50 cGy (200 cGy), 100 cGy (250 cGy/min), 150 cGy (300 cGy/min), 200 cGy (350 cGy/min), and 250 cGy (400 cGy) X-ray at an exposure time of 0.25, 0.4, 0.5, 0.57, 0.63 min, respectively, using a 6 MV photon beam linear accelerator (Elekta Synergy Platform). The thin film was positioned such that it was parallel to the X-ray beam. Secondary charged particle equilibrium was implemented using an additional PMMA build-up layer in front of the thin film sensor, as prescribed by ISO 4037-3<sup>[42]</sup>. The irradiation field size of the beam is  $5 \times 5$  cm<sup>2</sup>. After the setup, the researcher exits the Linacs bunker, and in accordance with a radiation safety policy, waits 60 seconds after the exposure before entering the bunker again. CCTV footage of the bunker and a monitor in the control room that shows the electrometer readings are used to capture the induced current. With regard to the applied voltages of 0, 1, 2, 3, 4, 5, and 6 V, each measurement was done independently.

#### 2.5 Test for dosimetric fading

The I-V characteristics of the thin film sensor were tested for fading. These measurements were performed for post-irradiation of the thin films to high-energy X-rays. The I-V characteristics measurement was carried out at an interval of 0, 1, 2, 15, and 30 days after irradiation respectively. At the end of the I-V characteristics measurements, the values of measured current were normalised to values of day 0 using Equation (1):

$$Normalisation = \frac{I_x}{I_{0x}} \times 100\%$$
(1)

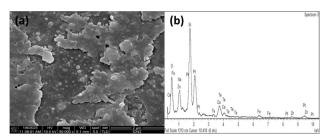
where  $I_x$  is the measured current at a given day for x voltage and  $I_{0x}$  is the measured current at day 0 for x voltage. The normalised values are presented in percentages.

# 3. Results and discussion

The FESEM images of  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film are shown in **Figure 2a**. The FESESM was used to image and identify the morphology of the film taken at 50,000× magnification. As we know, ZnO and TeO<sub>2</sub> have a hexagonal and tetragonal structure, respectively, and the FESEM image showed a non-uniform distribution of clustered structures. As

shown in Figure 2a, the TeO<sub>2</sub>-doped ZnO thin layers are stacked together. As a result, the nanoparticles within the film are seen agglomerated separately in the FESEM image of the film. The nanoparticles are non-uniform in shape, and the grains are non-crystalline and non-uniform. Similar non-uniform surface morphology was reported by Dobri et al. <sup>[43]</sup> for TeO<sub>2</sub> thin films, Shirpay and Bagheri <sup>[44]</sup> for MoTeO<sub>2</sub> binary thin films, Urfa et al. <sup>[45]</sup> for ZnO nanoparticles, and Park et al. <sup>[46]</sup> for TeO<sub>2</sub>-core/TiO<sub>2</sub>-shell nanowires. Although the samples in the current and previous studies have non-identical morphology and the data differed due to differences in sample preparation, method of deposition, the thickness of the film, chemical sample and post-deposition treatment condition <sup>[25,47,48]</sup>.

The chemical composition of the film was carried out with an energy-dispersive X-ray spectrometer (EDX) during FESEM analysis. Figure 2b shows the EDX spectrum of the  $(ZnO)_{0,2}(TeO_2)_{0,8}$  thin film nanostructure, revealing the existence of Te, Zn, and O peaks. After the quasi-quantitative determination of the EDX spectrum, the weight percentages of Te, Zn, and O were 8.09, 5.66, and 28.54, respectively, and the atomic percentages of Te (K), Zn (K), and O were 2.04, 2.78, and 57.32, respectively. It is demonstrated that the purity of the fabrication is high without the other residues such as Si, Ca, and Na derived from the soda-lime glass substrate and Pt buffer layer coating for the FESEM analysis. Traces of Fe in a negligible amount were seen, which might have found their way into the film during Pt coating for FESEM examination. The soda-lime glass substrate residues are relatively high in the EDX spectrum, not as constituents of the film but because of the thin nature of the film (a thickness of 375 nm), which enables the electron to interact with the glass substrate. It is also supposed that the ratio of Zn/O and  $Te/O_2$ is less than 1, compared with the perfect chemical stoichiometry of ZnO and TeO<sub>2</sub>. These results reveal that there is some oxygen vacancy in the ZnO/TeO<sub>2</sub> thin film. Similar observations have been reported by Silambarasan et al.<sup>[49]</sup> and Shanmugam et al.<sup>[50]</sup>.



**Figure 2.** (a) FESEM image of  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film (b) EDX spectrum of  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film.

The structural properties of the  $(ZnO)_{0.2}(TeO_2)_{0.8}$ thin film were studied by XRD measurement shown in Figure 3. The as-deposited thin film is non-crystalline (amorphous) in nature. The pattern shows a highly concentrated weak diffraction peak present at about 23.25° belonging to the diffraction pattern of the  $\alpha$ -phase TeO<sub>2</sub> ( $\alpha$ -TeO<sub>2</sub>, JCPDS card 78-1713) with preferred orientation along (110) direction. A similar XRD diffraction spectrum was reported by Jeong et al. <sup>[51]</sup> for the dual active layer of IGZO thin film transistor, Khan et al. [52] for as-deposited multilayer ZnO/TiO<sub>2</sub> thin films and Carotenuto et al.<sup>[53]</sup> for tellurium film. A weak intensive peak at 68.32° is seen which may be due to the wurtzite structure of ZnO (JCPDS card 043-0002) with preferred orientation along (201) direction as reported by Shatnawi et al. <sup>[54]</sup>. Both  $\alpha$ -TeO<sub>2</sub> and wurtzite ZnO phases were already present in the as-deposited (ZnO)<sub>0.2</sub>(TeO<sub>2</sub>)<sub>0.8</sub> thin film.

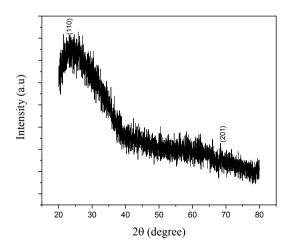
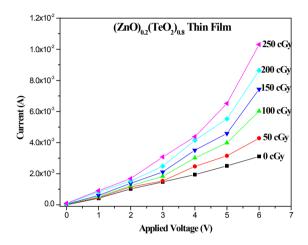


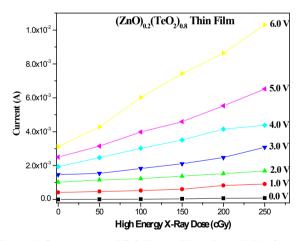
Figure 3. XRD pattern of (ZnO)<sub>0.2</sub>(TeO<sub>2</sub>)<sub>0.8</sub> thin film.

**Figure 4** shows typical current versus applied voltage plots for the  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor exposed to various X-ray doses. The dependence

of the current density at different voltages applied to the thin film structure is shown in **Figure 5**. Clearly, the current density has been found to increase linearly with the X-ray dose over a range of 0-250 cGy (**Figure 4**). The sensitivities of the  $(ZnO)_{0.2}(TeO_2)_{0.8}$ thin film device were calculated using current density versus dose plot and was found to be in the 0.37-0.94 mA/cm<sup>2</sup>/Gy range.



**Figure 4.** Current versus Voltage (I-V) plot at different X-ray Dose for  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film device.



**Figure 5.** Current versus High Energy X-ray Dose (I-D) plot at different applied Voltages for  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film device.

The current versus voltage (I-V) characteristics of the prepared thin films show an enhancement in the values of current under a forward-biassed condition. During the transmission of X-rays through thin films, defects are induced, resulting in the disordered structure of the film. This is attributed to the energy transfer from the X-ray radiation dose to the electrons that makes the electron move from the valence band to the conduction band, creating electron-hole pairs that increase the conductivity of the film <sup>[16]</sup>.

The test for fading of the I-V characteristics of the post-X-ray irradiation for days 0, 1, 2, 15, and 30 normalised to day 0 was analyzed. The mean relative readings obtained are in the order of day 0 > day 15 > day 30 > day 1 > day 2. Firstly, there was a decline in the induced current on days 1 (81.59%) and 2 (89.5%). An abrupt increase was observed on day 15 (99.20%) and a slight decrease on day 30 (93.30%). These abnormalities could be attributed to the introduction of ZnO into the lattice of TeO<sub>2</sub> in a small amount (20% ZnO). The semiconducting behaviour of the film indicates that the irradiation with high-energy X-ray radiation affects the microstructure of the film <sup>[38-42]</sup>.

The healing effect may also be responsible for the rise in current density <sup>[55]</sup>. There are usually some intrinsic flaws created during the film deposition. As a result of the interaction between X-ray radiation and the film, the film's structure will become disorganised throughout transmission through the film. The number of defects (induced plus residual intrinsic defects) is less than the number of intrinsic defects due to the recombination of defects, which lowers the thin film's resistivity and results in an increase in current <sup>[56]</sup>. At small doses, these thin films have a fine homogeneous grain structure without any large pores. The healing effect may also be used to understand how the critical dosage depends on the film thickness <sup>[48]</sup>. Therefore, compared to thinner films, larger films require substantially greater radiation doses for healing <sup>[29]</sup>.

Transparent conducting oxide thin layer conductivity control is still a difficult task <sup>[38,39]</sup>. The optical and electrical characteristics of semiconductor thin film devices can be strongly impacted by even very low concentrations of impurities (down to 10<sup>-14</sup> cm<sup>3</sup> or 0.01 ppm) and native point defects <sup>[57-59]</sup>. In order to effectively manage the conductivity in a thin film device, it is essential to comprehend the function of native point defects (such as vacancies, interstitials, and antisites), the integration of impurities, and the doping of two or more oxides. It has long been confirmed that oxygen vacancies or interstitials are what produce the unintended n-type conductivity in ZnO <sup>[60]</sup>. Pure ZnO thin films are not resistant to corrosive conditions; for instance, the adsorption of  $O_2$ reduces the film's electrical conductivity and alters the surface shape <sup>[61]</sup>. In order to make the ZnO system resistant to such changes, doping ZnO with TeO<sub>2</sub> has been examined in this work, leading to the development of an intriguing family of materials based on doped ZnO.

# 4. Conclusions

This work involves the synthesis, doping, film deposition, structural, morphological study, and the current-voltage characteristics during high energy X-ray exposure of a  $(ZnO)_{0.2}(TeO_2)_{0.8}$  thin film sensor prepared by the spray pyrolysis method. The XRD pattern revealed the non-crystalline (amorphous) nature of the film. The FESEM image shows non-uniform shapes of nanoparticles that are agglomerated separately. The current density has been found to increase linearly with the X-ray dose over a range of 0-250 cGy. The sensitivity of the film device was estimated and found to be in the range of 0.37-0.94 mA/cm<sup>2</sup>/Gy. There is a slight decline in the film conductivity over a period of 30 days after exposure to high-energy X-rays. The semiconducting behaviour of the film indicates that the irradiation with high-energy X-ray doses affects the microstructure of the film. These results are expected to be beneficial for fabricating cheap and practical X-ray detection applications.

# **Conflict of Interest**

There is no conflict of interest.

# Acknowledgement

The authors wish to thank the Nigeria government through the Tertiary Education Trust Fund (TET Fund) for funding this research study.

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